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High pressure Raman, optical absorption and resistivity study of SrCrO₄

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ABSTRACT: We have studied the electronic and vibrational properties of monazite-type SrCrO₄ under compression. The study extended the pressure range of previous studies from 26 to 58 GPa. The existence of two previously reported phase transitions has been confirmed at 9 and 14 GPa and two new phase transitions have been found at 35 GPa and 48 GPa. These transitions involve several changes in the vibrational and transport properties with the new high-pressure phases having a lower conductivity than the previously known phases. No evidence of chemical decomposition or metallization of SrCrO₄ has been detected. A tentative explanation for the reported observations is discussed.

Keywords: monazite, chromate, high pressure, phase transition, Raman, optical absorption, resistivity
1. Introduction

Photocatalytic materials which respond to ultra-violet and visible light have gained much attention during recent years. Among them, chromates of general formula ACrO$_4$, where A is a divalent metal (e.g. Sr, Ba, and Pb), have been found to have excellent optical-absorption properties, a high photocatalytic activity, and a band-gap energy near 2.5 eV. Such unique properties make ACrO$_4$ chromates useful not only in photocatalysis, but also in a wide range of applications, including photosensitization, photoluminescence, and scintillation.$^1$–$^4$

Strontium chromate (SrCrO$_4$) has the advantage against its Ba and Pb analogues of not being toxic to humans.

The crystal structure of SrCrO$_4$ at ambient conditions is well known, being isomorphous to the monazite structure (space group P2$_1$/n).$^5$ It consists of Sr atoms coordinated by nine oxygen atoms and distorted CrO$_4$ tetrahedral units. Its main physical properties at atmospheric conditions have been studied experimentally and theoretically.$^6$–$^8$ SrCrO$_4$ has been characterized as a semiconductor material with a band-gap energy of 2.45 eV.$^7$ The frequencies of the Raman-active phonon modes have been accurately determined.$^7$ Such information has contributed considerably to establishing the composition of late 19th-early 20th century oil paintings.$^9$

The effect of compression in the properties of SrCrO$_4$ has been recently studied up to 26 GPa.$^{10}$ In these studies, evidence of two phase transitions at 8–9 and 10–13 GPa has been found. The crystal structures of the high-pressure (HP) phases have been assigned to the tetragonal scheelite-type and monoclinic AgMnO$_4$-type structures. The influence of these transitions in the lattice vibrations and electronic band-structure has been also established.$^{10}$ The interest on studies comes from the fact that they have been shown to be an effective instrument for deepening the understanding of properties of oxides related to SrCrO$_4$.$^{11}$–$^{15}$ The extension of studies in SrCrO$_4$ towards higher pressures is of interest for several reasons: To search for new phase transitions, including possible pressure-induced amorphization$^{16}$ and metallization,$^{17}$ and the determination of possible effects of pressure on the chemical composition.$^{18}$ All these phenomena have been found to occur in oxides related to SrCrO$_4$ at pressures not far from 20 GPa.
Here, we present a HP Raman study of monazite-type SrCrO$_4$ up to 58 GPa. The study is combined with optical-absorption measurements up to 32 GPa and resistivity measurements up to 50 GPa. In the next sections we will describe the experimental methods and report and discuss the results of the present experiments. Evidence of two previously unknown phase transitions will be reported. The behaviour of SrCrO$_4$ under HP will be compared with that of related oxides.

2. Experimental details

Single crystals of SrCrO$_4$ were grown by solid-state reaction following the procedure described in Ref. 10. The high purity of the synthesized crystals was confirmed by energy-dispersive X-ray spectroscopy measurements carried out using a transmission-electron microscope operated at 200 KeV. The crystal structure was verified by powder X-ray diffraction using Cu K$_\alpha$ radiation. It was assigned to the well-known monazite-type structure, with unit-cell parameters: $a = 7.065(7)$ Å, $b = 7.376(7)$ Å, $c = 6.741(7)$ Å, and $\beta = 103.1(1)^\circ$, in good agreement with the literature.$^5,7$

For the HP Raman measurements, we used a diamond-anvil cell (DAC) equipped with 250 μm diameter diamond culets. Rhenium gaskets were indented to a thickness of 40 μm. The pressure chamber was a 100 μm diameter hole drilled in its center. Ruby chips were loaded together with the sample for pressure determination using the calibration reported by Dewaele et al.$^{19}$ N$_2$ was employed as a pressure-transmitting medium (PTM).$^{20}$ Special caution was taken during the DAC loading to avoid sample bridging between the anvils.$^{21,22}$ Three independent Raman experiments were carried out. Raman spectra were measured under compression to 58 GPa. At each pressure, Raman spectra were collected using a single grating HORIBA Jobin Yvon iHR320 spectrometer with a 1200 gr/mm and a nitrogen-cooled HORIBA Jobin Yvon Symphony CCD. The backscattering geometry was used with a 532 nm laser (spot size of the order of 1 μm). A 20x magnification objective was employed to focus the laser on sample and collect the Raman signal. The incident power on the sample (10 mW) was kept constant in the sample along the Raman measurements. The spectral resolution was better than 6.5 cm$^{-1}$ full width half maximum.$^{23}$ Raman spectra were analysed using MagicPlot Student 2.5.1 software. The background was subtracted and Lorentzian curves were fitted to peaks emanating from SrCrO$_4$. 


For optical-absorption measurements, the same DAC setup was employed except that argon\textsuperscript{24} was used as PTM instead of N\textsubscript{2}. The measurements were carried out using a confocal setup built using an Ocean Optics DH-2000 light-source, Cassegrain objectives, and a USB2000 UV–VIS–NIR spectrometer from Ocean Optics.\textsuperscript{25,26} The absorption spectra were computed at selected pressures from the transmittance spectra which were collected using the sample-in sample-out method.\textsuperscript{27,28} The pressure was determined using the ruby scale; the maximum pressure achieved was 32 GPa.

Resistivity measurements were carried out up to 50 GPa using a Almax-Easylab OmniDac gas-membrane-driven DAC. The pressure was calibrated using the fluorescence of ruby. Resistivity was measured using a designer anvil\textsuperscript{29} with a culet diameter of approximately 180 μm. A powdered sample was loaded directly into a stainless steel gasket with no insulating material separating the sample from the gasket. Resistance was measured with a Lakeshore Model 370 AC resistance bridge using a two-probe setup. From measurements across different combinations of the wires and previous experiments with the same anvils, it was estimated that the contact resistances never exceeded 10 % of the total resistance. The sample resistivity, $\rho$, is obtained using the equation $\rho = (\pi t R / \ln 2)$, Where $R$ is the resistance and $t$ is the estimated thickness of the sample. We have used the same procedure and experimental setup previously to obtain accurate results on the pressure dependence of the resistivity in PbCrO\textsubscript{4} and other semiconducting materials.\textsuperscript{30,31}

3. Results and discussion

a. Raman spectroscopy

The Raman spectrum of SrCrO\textsubscript{4} under compression has been studied in the past by our group up to 26 GPa.\textsuperscript{10} To avoid redundancies, we concentrate here on new results from 26 to 58 GPa. The Raman-active modes of SrCrO\textsubscript{4} are located in three isolated frequency regions in which modes have very different intensities. The strongest modes, which are associated with internal stretching vibrations of the CrO\textsubscript{4} tetrahedron, are located above 800 cm\textsuperscript{-1} (high-frequency region).\textsuperscript{7,10} Then there are modes from 290 to 550 cm\textsuperscript{-1} (intermediate frequency region), which originate from internal bending vibrations of the CrO\textsubscript{4} ion, and whose intensity is approximately five times weaker than that of the high-frequency modes. And finally there are a group of weak modes below 290 cm\textsuperscript{-1}(low-frequency region), which involve movements
of the Sr atoms and CrO$_4$ ions as rigid units. The clearest evidence of phase transitions and chemical decomposition in SrCrO$_4$ and other monazite-type chromates is expected to be detected in the 290 – 550 cm$^{-1}$ and 800 – 1100 cm$^{-1}$ regions.$^{10,32}$ In particular, the high-frequency phonons are very sensitive to coordination changes for the Cr atom. Both the number of modes and the frequencies of the modes could change as a consequence of a structural transition. Based upon this fact, we focused the study in two sections of the spectrum of SrCrO$_4$, the medium wave-number region from 290 to 550 cm$^{-1}$ (which we extended from 200 to 600 cm$^{-1}$) and the high wave-number region from 800 to 1100 cm$^{-1}$, using longer acquisition times in the first region.

In Fig. 1 we show a selection of Raman spectra measured under compression. Evidence of the first transition can be seen at 10.5 GPa. See for instance the changes in the left panel when comparing the spectra at 8.1 and 10.5 GPa and the new peak appearing near 850 cm$^{-1}$ (right panel). This Raman spectrum corresponds to the coexistence of the monazite and scheelite phases of SrCrO$_4$. The second transition (to AgMnO$_4$-type SrCrO$_4$) is illustrated by the spectrum measured at 18.3 GPa. Notice for instance the new peaks at low frequencies (left panel) and the differences between the spectra measured at 13.4 and 18.1 GPa in the right panel. These results confirm previously published results.$^{10}$ The first transition corresponds to the monazite-scheelite transition previously reported at 9 GPa$^{10}$, the second transition to the scheelite-AgMnO$_4$-type transition reported at 13 GPa$^{10}$ but taking place at slightly higher pressure (beyond 13.4 GPa) in this work. From this pressure up to 35.1 GPa we have only observed a gradual evolution with pressure of the Raman spectrum. At this pressure we have detected qualitative changes in the Raman spectrum. The changes are reversible and include the increase of the number of Raman modes detected and the decrease of the intensity of the Raman signal. The mentioned changes can be clearly seen by comparing the spectra at 28.1 and 35.1 GPa in Fig. 1. A possible explanation for this observation is the existence of a third phase transition (to a phase we will name as phase IV). Such hypothesis is supported by the results of our optical and resistivity experiments to be discussed in the next section. Since at 35.1 GPa the experimental conditions were not quasi-hydrostatic$^{20}$, there is a possibility that the phase transition at 35.1 GPa could be triggered by the presence of large strains.$^{33}$ The study in detail of such hypothesis is beyond the scope of the present work.
Upon further compression we did not observe any qualitative change in the Raman spectrum up to 48 GPa. However, the Raman signal intensity gradually decreases as pressure is increased until there is no detectable Raman signal past 48 GPa and the sample appeared black. We consider this observation as the evidence of the existence of a possible fourth phase transition to a phase we will name as phase V. The blackening of the sample starts at 43 GPa, is gradual and can be caused by precursor defects of the fourth phase transition. This phenomenon is reversible, albeit with significant hysteresis (the Raman spectrum is not recovered until 26 GPa upon decompression). However permanent cracks are induced in the crystal, which remain on pressure release. This is typical of a first-order phase transition, but is also observed following pressure-induced amorphization (defined as the loss of long-range order). In Fig. 2 we show pictures of the sample to illustrate it.

We would like to now discuss the possible decomposition of SrCrO$_4$ under compression. Pressure-induced decomposition is expected to occur when the decomposition products occupy less volume than any polymorph of the parent material. Usually this phenomenon is irreversible and leads to the appearance of Raman modes that cannot be assigned to the parent material. In our case, no irreversible changes have been detected. The Raman spectrum of SrCrO$_4$ is recovered upon decompression with no additional modes present, ruling out the possible decomposition of SrCrO$_4$. Another phenomenon that should be considered is pressure induced amorphization, which usually occurs by the kinetic hindrance of a phase transition. However, our experimental findings indicate that SrCrO$_4$ does not undergo pressure induced amorphization. In oxides related to monazite, pressure-induced amorphization usually induces a precipitous decline in Raman intensity and the appearance of diffuse bands which remain upon pressure release instead of the original Raman peaks. In our case, the Raman signal decreases below the detection limit at 48 GPa, however, the phenomenon is fully reversible, so it cannot be assigned to amorphization. Notice that the recrystallization of an amorphous solid at room temperature is very unusual, being only reported up to now in molecular compounds such as SiH$_4$. This phenomenon is related to the bond directionality of these compounds, which differs from that of SrCrO$_4$. Additionally, the typical amorphous Raman signal has not been detected in our measurements. We believe that the loss of the Raman signal is related to the occurrence of the fourth phase transition. In the phase found beyond 48 GPa, the formation of a large population of cracks in the crystal,
described above, triggers a ‘disorder-induced’ decrease in intensity of the Raman scattering.\textsuperscript{41} Such an explanation should be confirmed by future studies. Of particular relevance would be the performance of HP-XRD studies using He as pressure-medium, HP-HT XRD measurements, and X-ray absorption studies at Sr k-edge to explore possible coordination changes associated with the fourth phase transition.

We will now discuss the mode assignment of Raman modes in the AgMnO\textsubscript{4}-type phase and the pressure dependence of the modes. We will focus only on bending and stretching modes of the CrO\textsubscript{4} tetrahedron (i.e. those with frequencies > 300 cm\textsuperscript{-1}). According to the crystal symmetry, this phase has thirty-six Raman-active phonons ($\Gamma = 18 A_g + 18 B_g$).\textsuperscript{10} Following density-functional theory calculations\textsuperscript{10} eight of these modes correspond to internal stretching modes of the CrO\textsubscript{4} tetrahedron and ten to bending vibrations of the same octahedron.\textsuperscript{10} The stretching (bending) vibrations are in the high-frequency (intermediate-frequency) region of the Raman spectrum. However, in previous measurements\textsuperscript{10} ten modes (instead of eight) were reported in the high-frequency region and eight modes (instead of ten) in the intermediate frequency region. Here we have found the expected number of modes in each of the frequency regions of the Raman spectrum. The results from present and previous\textsuperscript{10} experiments are compared in Table 1. The agreement in frequencies is quite good. The two modes not detected in the previous study\textsuperscript{10} are the $A_g$ modes with frequencies 340 and 375 cm\textsuperscript{-1} at 11.7 GPa, which are two weak modes. The two extra modes detected previously in the high-frequency region are the two weakest modes which probably correspond to Raman overtones.\textsuperscript{10} Their frequencies are 927 and 1002 cm\textsuperscript{-1} at 11.7 GPa, and not those previously proposed (851 and 891 cm\textsuperscript{-1}). The frequencies we propose here for the overtones, within the accuracy of measurements, agree with the sums of the frequencies of two lower frequencies $A_g$ modes\textsuperscript{10}; (852 + 78) cm\textsuperscript{-1} and (918 + 82) cm\textsuperscript{-1}, respectively. The pressure coefficients obtained for the overtones differ by 30\% from the sum of the respective pressure coefficients of the Raman modes involved in the overtones. The reason for this is possibly the very low intensities of the overtones, and consequent inaccuracy in the determination of their pressure dependence.

Regarding the pressure dependence of the Raman frequencies, the obtained results are shown in Fig. 3. There it can be seen that most of the modes follow a non-linear behaviour, which can be described by a quadratic function. The results of the quadratic fits are shown in
Table 1. In the table it can be seen that the high-frequency stretching modes are those with the largest pressure coefficients. In particular, the mode most sensitive to pressure is the $A_g$ mode with frequency 918 cm$^{-1}$ at 11.7 GPa. The fact that different modes have different pressure dependence leads to several modes gradually merging under compression. This is for instance the case of the $B_g$ and $A_g$ modes with frequencies 411 and 419 cm$^{-1}$ at 11.7 GPa. They are represented by open blue symbols in Fig. 3. Under compression the frequency of the $B_g$ mode increases faster than that of the $A_g$ mode, reducing the frequency difference between the two modes. The modes merge and cannot be distinguished above 25 GPa. Another fact to highlight is the existence of anti-crossing modes. These are the $B_g$ modes with frequencies 353 and 371 cm$^{-1}$ at 11.7 GPa, which are shown with open red symbols to facilitate the identification of the anti-crossing behaviour. A final remark, is that the quadratic coefficient ($\omega_2$) in the function that describes the pressure dependence of Raman modes is negative in most modes (see Table 1), with the pressure dependence having a negative curvature, decreasing the slope of the curves as pressure increases. This is particularly noticeable for the high-frequency modes; in all of them $\omega_2 < 0$.

We will now comment on the Raman spectrum of phase IV. Interestingly for this phase we also observed eighteen Raman-active modes in the frequency region covered by our measurements. The frequency of these modes and their pressure dependence are summarized in Table 2. The pressure dependence of the modes is shown in Fig. 3. There it can be seen that all modes harden under compression. Raman experiments do not allow the determination of the crystal structure of phase IV (we hope our study will trigger the X-ray diffraction experiments needed for it) but they can give some hints which can guide the search for the HP structure of phase IV. The fact that the AgMnO$_4$-type structure of SrCrO$_4$ and phase IV have the same number of modes with a similar frequency distribution is indicative that there is little (or no) change in the coordination number of Cr and Sr at the phase transition. The fact that the high- and intermediate-frequency modes cover a wider frequency range in phase IV than in the AgMnO$_4$-type phase suggest a distortion of the CrO$_4$ octahedron. The shift of the highest-frequency mode towards high frequency after the phase transition is suggesting a shortening of the Cr-O bond after the phase transition. This is expected to enhance the bond strength causing an increase of the Raman stretching frequency. The fact that there are no coordination number changes at the AgMnO$_4$-type to phase IV transition
suggest that the crystal structure of phase IV should be related to that of the AgMnO$_4$-type phase, the transition probably being displacive, which is consistent with the fact that it is reversible.$^{44}$ A literature search indicates that there are two structures with similarities to the AgMnO$_4$ structure in compounds related to SrCrO$_4$.$^{45,46}$ These are the barite (space group Pnma) and another orthorhombic structure described by space group P2$_1$2$_1$2$_1$. The first structure has the same number of Raman modes as the AgMnO$_4$-type structure (11 Ag + 7 B1g + 11 B2g + 7 B3g), however, the second has sixty-nine Raman active modes (18 A + 17 B1 + 17 B2 + 17 B3), which exclude it as a potential HP phase for SrCrO$_4$. Therefore, amongst the two candidate structures, only barite (a high-symmetry version of AgMnO$_4$) has a Raman spectrum consistent with our finding for phase IV of SrCrO$_4$. A definitive identification of the crystal structure of it should be obtained in future HP X-ray diffraction experiments.

Before discussing the optical and resistivity measurements, we would like to comment on the pressure dependence of Raman modes in phase IV. The results obtained are summarized in Fig. 3. In the intermediate-frequency region the modes can be followed up to 39 GPa before they vanished. In the high-frequency region (the most intense modes) the modes can be followed up to 45 GPa. In both regions the pressure dependence of the frequencies is nearly linear. The frequencies at 35.1 GPa and the linear pressure coefficients ($\omega_1$) are shown in Table 2. Notice that most modes have a smaller pressure coefficient than in the AgMnO$_4$-type phase. The mode with the largest pressure coefficient is the mode with frequency 516 cm$^{-1}$. Interestingly, in phase IV not all the modes harden under compression. There are two modes with negative pressure coefficients (see Table 2). The presence of such modes is usually a hint of structural instabilities$^{47}$, which is consistent with the evidence of a pressure-induced phase transition we found at 45 GPa.

b. Optical absorption

We will now discuss the results from optical-absorption experiments. Fig. 4 shows a selection of spectra measured at different pressures. The reported measurements extended the pressure range covered by previous studies$^{10}$ from 14.5 to 30.7 GPa. In the previous studies, it has been determined that SrCrO$_4$ is an indirect band gap material. Our results fully agree with those previously reported$^{10}$ up to 14.5 GPa, following the absorption spectra the quadratic dependence in energy expected for an indirect gap. From 14.5 to 30.7 GPa there
are no noticeable changes in the shape of the absorption spectrum (see Fig. 4); consequently, it can be assumed that SrCrO$_4$ is an indirect band-gap material up to 30.7 GPa.

We will discuss now the pressure dependence of the band-gap energy ($E_g$) in detail. We found that the absorption edge gradually blue-shifts from ambient pressure, where the $E_g$ is 2.45 eV to 8.3 GPa where $E_g = 2.6$ eV. At 8.5 eV an abrupt decrease of the band-gap takes place, red-shifting the absorption spectrum by 0.2 eV. The absorption spectrum of SrCrO$_4$ gradually shifts to higher energies upon further compression. From the absorption spectra we determined the pressure dependence of $E_g$, with an error of ± 0.02 eV. The results are summarized in Fig. 5 and compared with previously published results. The agreement is quite good in the pressure range where comparison is possible. In the figure, black, red, and blue symbols are used to represent results from the three known phases of SrCrO$_4$; i.e. monazite, scheelite, and AgMnO$_4$-type. The first transition occurs at 8.5 GPa as the band gap collapses. There is a change in the evolution of band-gap energy at 8.5 GPa, in agreement with the occurrence of the second transition which was detected by Raman and XRD measurements. Beyond this pressure, $E_g$ evolves linearly with pressure up to 30.7 GPa, with no evidence of phase transitions. Interestingly in the AgMnO$_4$-type phase near 15 GPa $E_g$ reaches the same value that the monazite-phase exhibits at ambient pressure. At 30.7 GPa the value of $E_g$ is 2.5 eV. Upon further compression, at 32.3 GPa the crystal breaks into multiple domains which prevent the performance of absorption measurements because light becomes diffused instead of being transmitted through the sample. We suspect that the breaking of the crystal into domains is a consequence of precursor effects of the phase transition to phase IV detected by Raman experiments at 35 GPa.

We will discuss now the pressure evolution of $E_g$. From Figs. 4 and 5 it is evident that SrCrO$_4$ is a semiconductor material within the pressure range covered by experiments. In contrast with PbCrO$_4$, the band gap of SrCrO$_4$ opens under compression in the monazite phase.$^{30}$ In accordance with band structure calculations$^{14}$, the opening of $E_g$ in SrCrO$_4$ is caused by the repulsion between the Cr and O states that contribute to the top of the valence band and the bottom of the conduction band. For the case of PbCrO$_4$, the contribution of Pb states to the valence and conduction bands is what favours the closing of $E_g$ under compression.$^{30}$ The 0.2 eV collapse of $E_g$ observed at the first transition (8.5 GPa) is a
consequence of the change of Cr–O bond angles and distances at the monazite-scheelite transition. Notice that the transition modifies the global symmetry of the crystal and the Cr–O bond distances. These changes are reflected in the electronic structure of SrCrO₄, which trigger the small closure of E₈ found in the experiments. On the other hand, the change of the slope observed in the pressure dependence of E₈ at the scheelite-AgMnO₄-type transition (10.5 GPa) is a consequence of the change in the compressibility associated to the transition. Finally, the opening of E₈ that takes place under compression in the two HP phases is a consequence of the volume reduction of the CrO₄ tetrahedron (as in the monazite phase) which increases the crystal field splitting between bonding and anti-bonding states causing the opening of the band gap.

c. Resistivity

Fig. 6 presents the results of the high-pressure electrical resistivity measurements. They have been plotted on a semi-log plot. There is a clear trend of decreasing resistivity under compression, with an overall change of more than one order of magnitude over the measured pressure range. The resistance of the sample was too high to reliably measure for pressures below 16 GPa. This is similar to the findings for PbCrO₄. After the phase transition to the AgMnO₄-type phase the resistivity becomes detectable. A change in the slope of the resistivity curve occurs near 35 GPa, which is followed by a plateau up to 42 GPa. The slope change at 35 GPa is likely connected with the previously discussed structural transition to phase IV. A second change in the slope of resistivity is present at 42 GPa, which is agreement with the transition to phase V that causes the blackening of the crystal. The data (ρ > 80 Ω.cm) suggest that SrCrO₄ remains non-metallic to at least 50 GPa.

The large resistivity observed below 16 GPa is consistent with an intrinsic semiconducting behaviour of SrCrO₄. At 300 K, assuming the same electron and hole mobility as in Si, a resistivity larger than 10⁷ Ω.cm is obtained for monazite-type SrCrO₄, which explains why the resistivity was too high to be accurately measured below 16 GPa. Beyond this pressure the most likely reason for explaining the decrease of the resistivity to detectable levels is the creation of oxygen vacancies which could act as donor levels transferring electrons to the conduction band. The presence of such vacancies is typical of chromate compounds. Theoretical calculations on the related compound BiVO₄ have shown that the
formation energy of oxygen vacancies can be provided by a hydrostatic pressure of 7 GPa. Therefore, it is not impossible to create oxygen vacancies in SrCrO$_4$ at 16 GPa. These vacancies will act as donor impurities and provide n-type doping to SrCrO$_4$, transforming it into an extrinsic semiconductor. As a consequence, the carrier concentration will be increased leading to the increased conductivity, as found in oxygen-deficient NdCrO$_4$, LaCrO$_4$, and other compounds related to SrCrO$_4$.

Regarding the decrease of resistivity from 16 to 35 GPa (AgMnO$_4$-type phase), it cannot correspond to the changes in the band gap. In fact, $E_g$ increases under compression as discussed in the previous section, which as a first approximation should produce an increase of resistivity. There are several reasons that could explain the decrease of resistivity. One is the enhancement of contacts between grains and the triggering of percolation effects due to compression, two facts that has been recently shown to favour the decrease of resistivity in compressed powder samples like the one here used. A second possible reason is that under compression the donors that contribute free electrons to the conduction band transform from deep to shallow leading to an increase of the carrier concentration and the consequent decrease of resistivity. Regarding the plateau observed in the resistivity from 35 to 42 GPa and the slope change found at 42 GPa, we can only state that they indicate a change in the electrical properties of SrCrO$_4$ which is consistent with the two new phase transitions found in Raman experiments. The value of the resistivity found at the highest pressure 80 Ω.cm (at 50 GPa) is an order of magnitude larger than the expected resistivity of a metal. Therefore, this excludes the possibility that the sample blackening described before could be caused by metallization. Taking this into account, a possible explanation for SrCrO$_4$ becoming black would be the formation of extrinsic defects - a phenomenon observed under high-pressure in InSe and GaSe at pressures much lower than the metallization pressure.

4. Concluding remarks

In this work we reported a high pressure experimental study of SrCrO$_4$. Raman, optical absorption and resistivity measurements were carried out expanding on the previous work performed by exploring a much higher pressure range. Two further transitions at 35 and 48 GPa were detected in addition to the previously observed transitions at 9 and 14 GPa. The phase V formed at 48 GPa is not fully characterized; we believe that it is not an amorphous
phase, on account of the reversibility of the transition, but that a high level of structural disorder may be present. The pressure dependence of Raman active modes has been discussed for two of the HP phases. In addition, optical-absorption experiments allowed the characterization of the band gap of SrCrO$_4$ up to 30.7 GPa (doubling the pressure range covered previously$^{10}$) showing that SrCrO$_4$ is a semiconductor. Important contributions in this work are the resistivity measurements and the confirmation that SrCrO$_4$ remains a non-metallic material at 50 GPa. The results shown further enhance our understanding of the high-pressure behaviour of monazite-type oxides and ternary compounds and can have implications in geochronology and petrology since monazite-type oxides are key accessory mineral for metamorphic rocks.$^{60}$

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**Author Contributions**

The manuscript was written through contributions of all authors. All authors have given approval to the final version of the manuscript.

**Notes**

The authors declare no competing financial interests.

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Figure 1: Raman spectra measured in SrCrO$_4$ at selected pressures. The left and right panels correspond to the two frequency regions described in the text. The magnification of spectra measured at 35.1 and 39.6 GPa is indicated in the figures.

Figure 2: Images of SrCrO$_4$ crystal under compression. (a) SrCrO$_4$ at ambient pressure visible in yellow-orange colour. (b) SrCrO$_4$ at 32.3 GPa experiencing a colour change from yellow to orange at the centre. (c) Colour change to black at 43.1 GPa. (d) SrCrO$_4$ at 25.7 GPa on decompression, experiencing a colour change from black to orange, but having permanent cracks.
Figure 3: Pressure dependence of Raman modes in SrCrO$_4$ from 11 to 45 GPa. Black solid circles are from Ref. 10. Green solid circles are from Ref. 10 and correspond to overtones (see...
text). Open circles represent results from the present results for the AgMnO$_4$-type phase. We represent in blue one $A_g$ and one $B_g$ mode which merges under compression. We show in red the anti-crossing of two $B_g$. Results for the new HP phase observed beyond 35 GPa are represent with diamonds.

**Figure 4:** Optical-absorption spectra measured at different pressures. Results for different phases are shown in different colours. Experiments at different pressures are shown using different style of lines. Pressures are indicated in the onset.
Figure 5: Pressure dependence of the band-gap energy (Eg). Solid symbols are from previous experiments. Empty symbols are from present experiments. Different colours are used for different phases.
Figure 6: Resistivity versus pressure on a semi-log plot. There are two changes in the slope, one near 35 GPa and the second at 42 GPa.
References


(5) Effenberger, H.; Pertlik, F. Four monazite type structures: comparison of SrCrO$_4$, SrSeO$_4$, PbCrO$_4$ (crocoite), and PbSeO$_4$. Zeitschrift für Kristallographie **1986**, *176*, 75-83.


(52) Aoki, Y.; Konno, H.; Tachikawa, H. The electronic and magnetic properties of LaCrO$_4$ and Nd$_{1-x}$Ca$_x$CrO$_4$ (x = 0–0.2) and the conduction mechanism. J. Mater. Chem. 2001, 11, 1214-1221.


Table 1: Wave numbers ($\omega_0$) determined at 11.7 GPa for Raman modes of AgMnO$_4$-type SrCrO$_4$ (in cm$^{-1}$) including mode assignment. The pressure dependence of the frequency of each mode is described by $\omega(P) = \omega_0 + \omega_1 \Delta P + \omega_2 \Delta P^2$, with $\Delta P = P - 11.7$ GPa. The wave number determined previously$^{10}$ are included for comparison. The overtones (see text) are also included.

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Table 2: Wave numbers ($\omega_0$) determined at 35.1 GPa for Raman modes of new phase IV of SrCrO$_4$ (in cm$^{-1}$). The pressure dependence of the frequency of each mode is described by $\omega(P) = \omega_0 + \omega_1 \Delta P$, with $\Delta P = P - 35.1$ GPa. One the left (right) the bending (stretching) modes are shown.

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HP phase transitions in SrCrO$_4$
Synopsis

Monazite-type SrCrO$_4$ has been studied at high pressure. We observe two new high pressure phases, characterized in terms of vibrational properties, optical absorption, and conductivity. The conductivity of the sample steadily decreased upon pressure increase throughout. We detected no direct evidence for chemical decomposition or metallization of SrCrO$_4$, although the structure of the highest pressure phase has yet to be determined.